

# Al-Mg-Ni (Aluminum-Magnesium-Nickel)

V. Raghavan

The early work on this system reviewed by [1993Pri] presented a vertical section along the  $\text{Mg}_2\text{Al}_3$ - $\text{NiAl}_3$  join and a liquidus projection in the Al-rich region. Recently, [2008He] determined two isothermal sections for this system at 700 and 427 °C, which depict the ternary compound  $\text{Ni}_2\text{Mg}_3\text{Al}$  (denoted  $\tau$  here).

## Binary Systems

The Al-Mg phase diagram [1998Lia] has the following intermediate phases:  $\text{Mg}_2\text{Al}_3$  (cubic, labeled  $\beta$ ), R or  $\varepsilon$  (rhombohedral), and  $\text{Mg}_{17}\text{Al}_{12}$  ( $A_{12}$ ,  $\alpha\text{Mn}$ -type cubic, denoted  $\gamma$ ). The Al-Ni phase diagram [1993Oka] shows five intermediate phases:  $\text{NiAl}_3$  ( $D0_{11}$ , Fe<sub>3</sub>C-type orthorhombic),  $\text{Ni}_2\text{Al}_3$  ( $D5_{13}$ -type hexagonal, denoted  $\delta$ ),  $\text{NiAl}$  ( $B2$ , CsCl-type cubic, denoted  $\beta$ ),  $\text{Ni}_5\text{Al}_3$  ( $\text{Ga}_3\text{Pt}_5$ -type orthorhombic), and  $\text{Ni}_3\text{Al}$  ( $L1_2$ , AuCu<sub>3</sub>-type cubic, denoted  $\gamma'$ ). The Mg-Ni phase diagram [Massalski2] depicts two intermediate compounds:  $\text{Mg}_2\text{Ni}$  (hexagonal, space group  $P6_222$ ) and  $\text{MgNi}_2$  (C36-type hexagonal).

## Ternary Isothermal Sections

With starting metal powders of 99.5% Al, 99.5% Mg, and 99.9% Ni, [2008He] sintered 14 ternary powder compacts, which were given a final anneal at 700 °C for 20 d or at 427 °C for 25 d and quenched in water. The phase equilibria were studied with x-ray powder diffraction, optical and scanning electron microscopy. The local phase compositions were measured by energy dispersive x-ray spectroscopy or electron probe microanalysis and listed. The isothermal sections at 700 and 427 °C constructed by [2008He] are redrawn in Fig. 1 and 2 to agree with the accepted binary data. The dotted phase boundaries are tentative, as no experimental measurements were done in the region, where liquid or (Ni) is present. The ternary phase  $\tau$  ( $\text{Ni}_2\text{Mg}_3\text{Al}$ ;  $\text{NiTi}_2$ -type cubic, with  $a = 1.15742$  nm) is present at the stoichiometric composition at 700 °C. At 427 °C, it has a small homogeneity region with a lattice parameter range of  $a = 1.15563$ - $1.15733$  nm. At 700 °C (Fig. 1), NiAl dissolves ~17 at.% Mg and  $\text{MgNi}_2$  dissolves ~10 at.% Al. At 427 °C (Fig. 2), NiAl and  $\text{Ni}_2\text{Al}_3$  dissolve ~16 and 2.5 at.% Mg [2008He].  $\text{MgNi}_2$  dissolves ~10 at.% Al. The solubility of Mg in NiAl indicated in Fig. 1 and 2 is, however, lower than

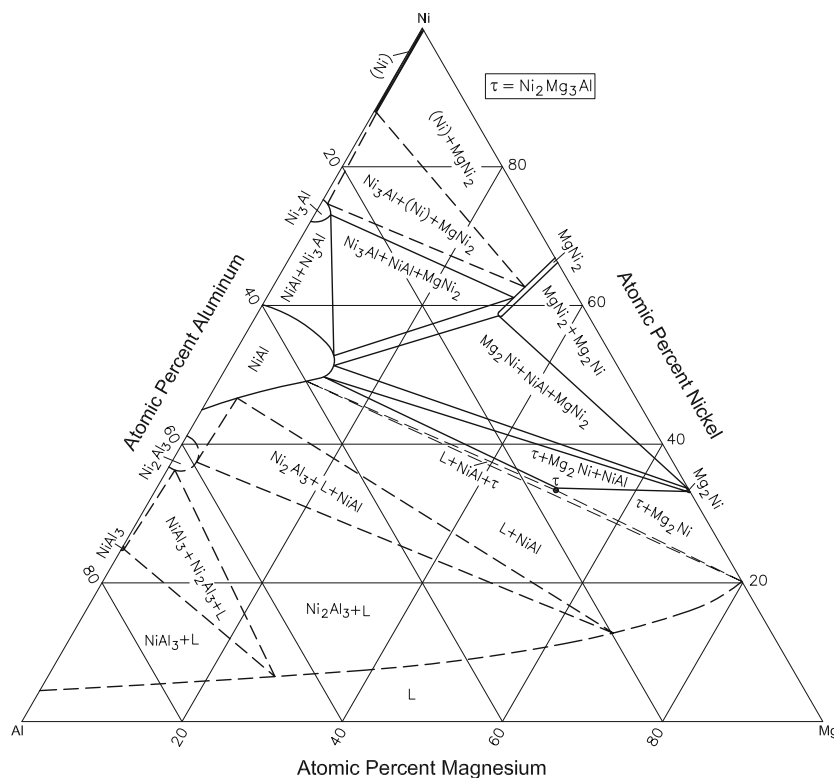
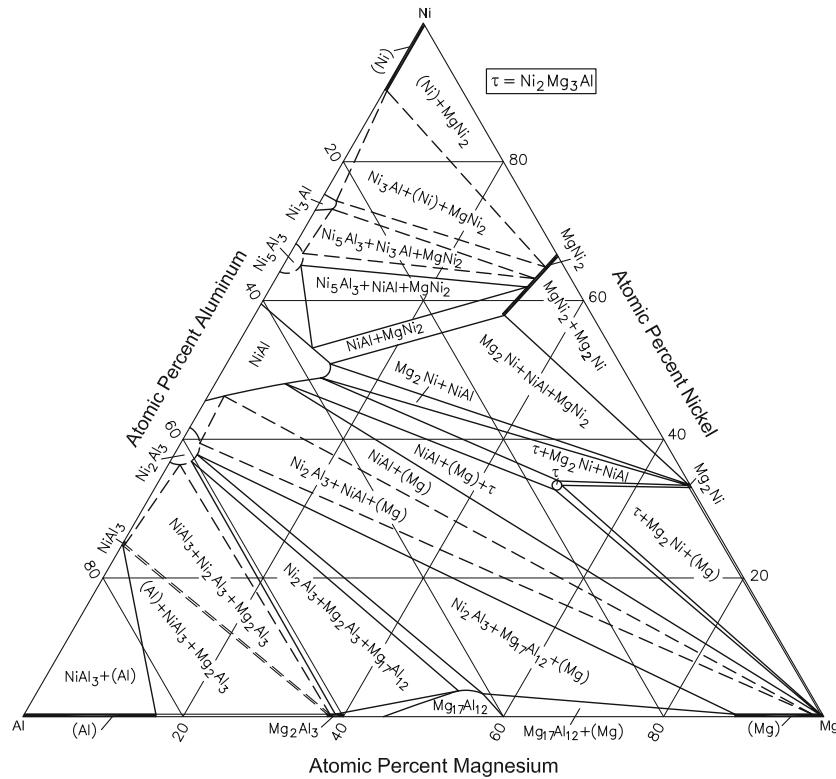


Fig. 1 Al-Mg-Ni isothermal section at 700 °C [2008He]



**Fig. 2** Al-Mg-Ni isothermal section at 427 °C [2008He]

16-17 at.% and is as given in the figures of [2008He]. In the samples annealed at 427 °C, two weak diffraction peaks were seen, which were indexed as MgO. This oxygen contamination was attributed to residual oxygen or moisture in the samples during processing.

## References

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